KINETICS OF AQUEOUS CORROSION

Electrode Kinetics;

* $M - ne^- \rightarrow M^{n+}$

  One mole of metal gives out $n N_A$ electrons which is a charge of $n N_A e$

  $\therefore I = \dfrac{dQ}{dt} = n N_A e\left(\dfrac{dN}{dt}\right)$ where N is the moles of material

  $\Rightarrow I = n F \left(-\dfrac{dN}{dt}\right)$

* Current density $i = \dfrac{I}{A} = \dfrac{nF}{A}\left(-\dfrac{dN}{dt}\right)$ where A = electrode area

Polarisation;

* For a half-cell reaction at one electrode; Ox. Species + $ne^- \leftrightarrow$ Red. Specie
* At equilibrium the rates of reaction forwards and backwards are
  equal $\therefore i_a = i_c = i_o$ (exchange current density). can't obtain directly
* i_o depends on surface chemistry so it is different for different metals, because
  $2H^+ + 2e^- \rightarrow H_2$ involves e^- transfer; H atom diffusion, and formation of H_2.
* In the above condition the electrode potential is $E_o^{\bullet}$ which is the standard
  E adjusted according to the Nernst equation.
* Polarisation is difference between actual E and $E_o^{\bullet}$;

  $$\eta = E - E_o \quad \text{so} \quad \eta > 0, E > E_o \text{ so this will cause } i_a > i_c$$
  $$\eta < 0, E < E_o \text{ which will cause } i_c > i_a$$

Activation Polarisation;

* When chemical reaction is the slowest step in the corrosion mechanism, the activation
  energy of the reaction is the most sig. step (activation polarisation).

* At eq. $i_o = i_a = i_c = A_o \exp\left(\dfrac{-\Delta G^*}{RT}\right)$

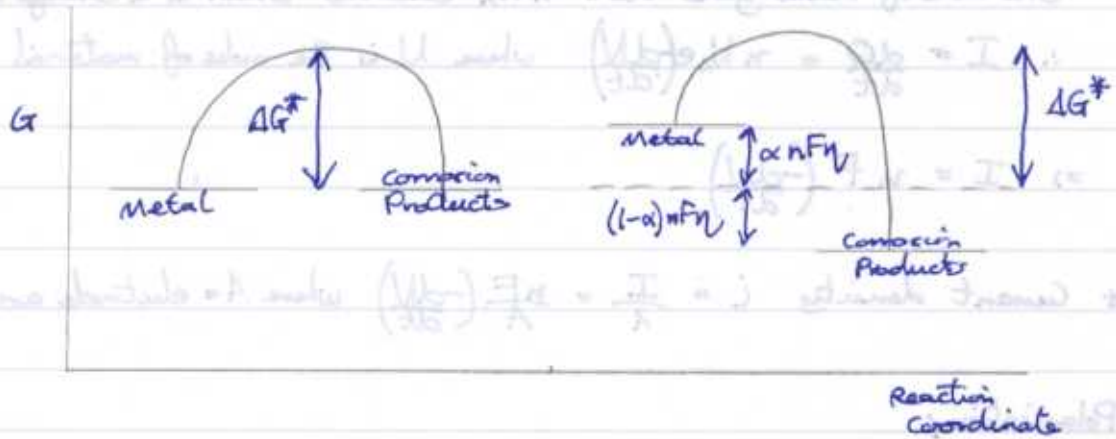 Now $\Delta G_R = -nF E_{cell}$ $\therefore \Delta G = nF\eta$

* As $i = i_a - i_c$;

$$i = A_o \exp\left(-\frac{(\Delta G^* - \alpha n F \eta)}{RT}\right) - A_o \exp\left(-\frac{(\Delta G^* + (1-\alpha)n F \eta)}{RT}\right)$$

$$\Rightarrow i = i_o\left[\exp\left(\frac{\alpha n F \eta}{RT}\right) - \exp\left(\frac{-(1-\alpha)n F \eta}{RT}\right)\right] \quad \text{Butler - Volmer}$$
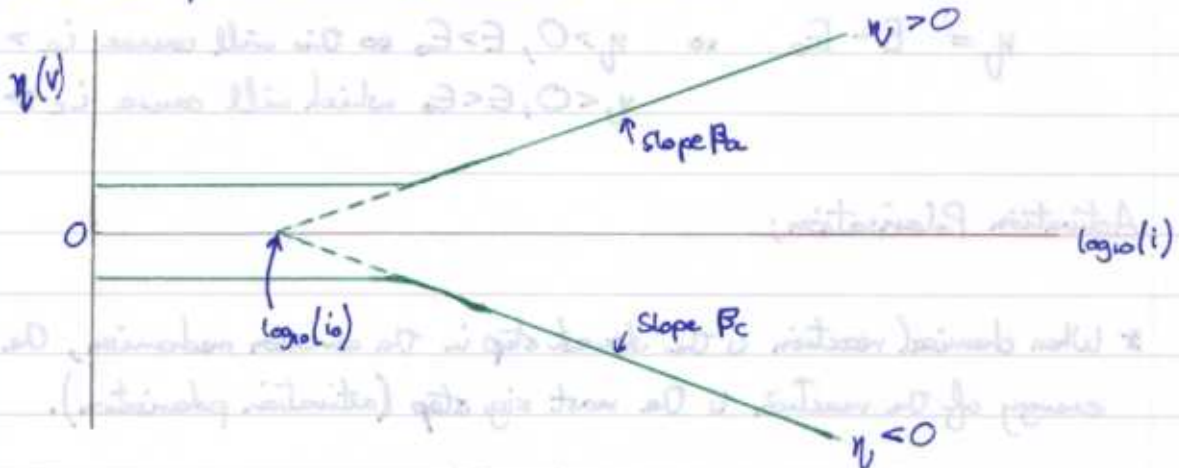

* If $\eta > 50mV$, then $i_a >> i_c$ so ;

$$\eta = \beta_a . \log\left(\frac{i}{i_o}\right) \quad \text{Tafel Eq} \text{ (for anodic reaction)}$$

(with $\beta_a = \frac{2.303 RT}{\alpha n F}$)

* If $\eta < -50mV$, then $i_c >> i_a$ so ;

$$\eta = \beta_c . \log\left(\frac{i}{i_o}\right) \quad \text{Tafel Eq} \text{ (for cathodic reaction)}$$

(with $\beta_c = \frac{-2.303 RT}{(1-\alpha)n F}$)

* At low polarisation behaviour is linear.

EVANS DIAGRAM → combines; thermodynamic and kinetic info

Mixed Potential Theory (2 half-cell reactions):
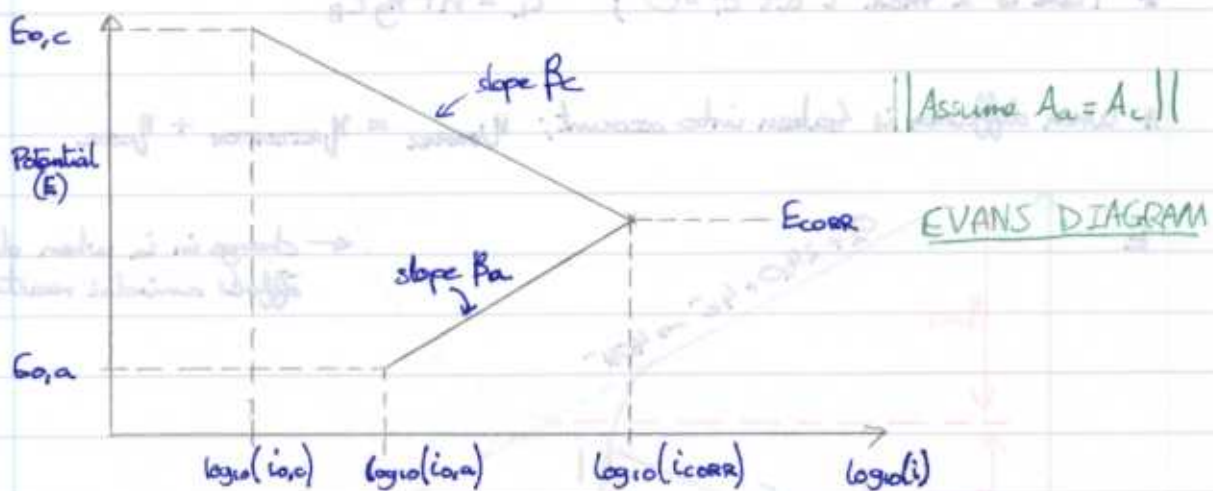
* Due to differences in microstructure 2 half-cell reaction can occur on the same surface.

* e.g. $M \rightarrow M^{a+} + 2e^-$ (anode) ⎫ Ignore back-reactions
  $2H^+ + 2e^- \rightarrow H_2$ (cathodic) ⎭

* The difference in E between the above reactions causes a flow of e^-. This will reduce the potential of the cathode and increase that of the anode.

* Charge conservation applies; I cell $= i_a A_a = i_c A_c$ where i_a, i_c depend on the potential of the metal.

* ⎡FREE
  ⎣MIXED CORROSION POTENTIAL; potential at which charge conservation is obeyed, in the absence of an external current source.

EVANS DIAGRAM

‖Assume $A_a = A_c$‖

slope β_c

slope β_a

E_{corr}

$E_{0,c}$, $E_{0,a}$, Potential (E)

$\log_{10}(i_{0,c})$, $\log_{10}(i_{0,a})$, $\log_{10}(i_{corr})$, $\log_{10}(i)$

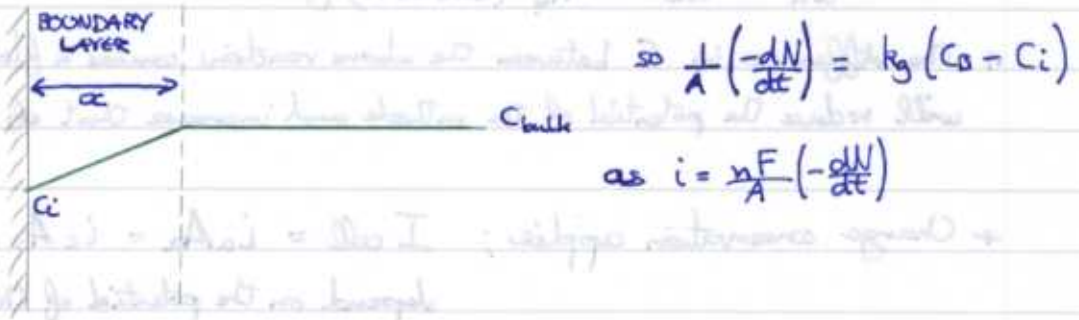* $E_{0,c}$ and $E_{0,a}$ are standard electrode potentials adjusted by Nernst Equation.
* Use above diagram and Tafel equations to find E_{corr}.
* i_{corr} depends on — electrode potentials
          — exchange current densities (which depend on surface chemistry)
          — Tafel parameters
          — external applied potentials.
* Due to charge conservation, corrosion will be faster if $A_a \ll A_c$
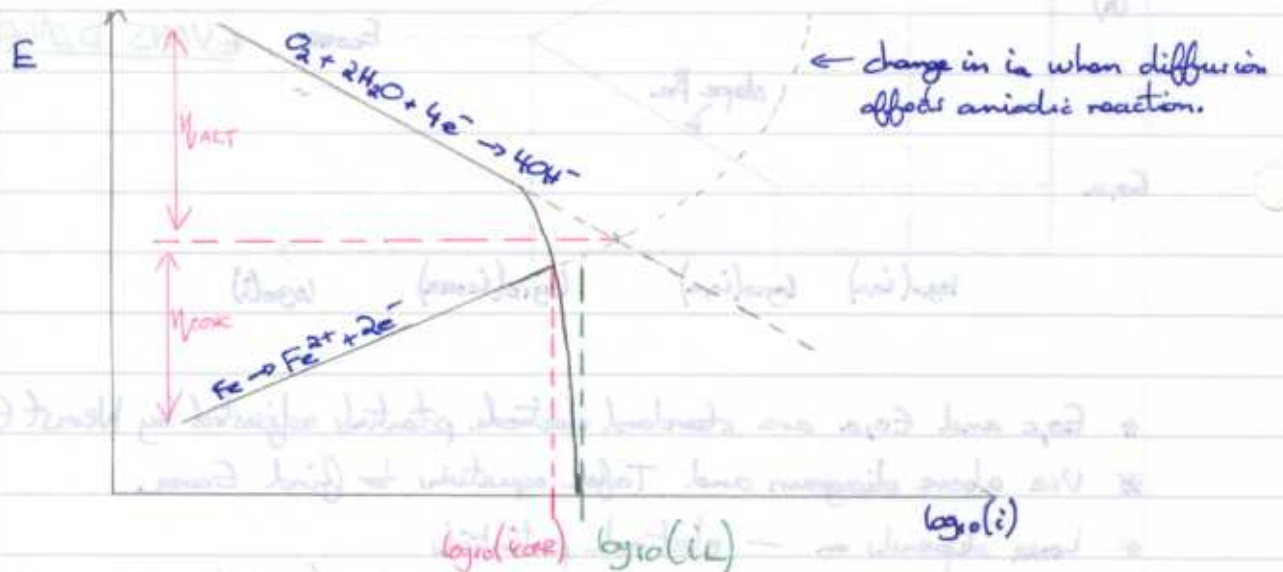
Diffusional Limitations; (Concentration polarisation)

* Diffusion of species to and from the electrode surface may be the rate determining step, not activation polarisation, particularly for reactions involving transport of GASes to the CATHODE.

BOUNDARY LAYER

x

C_{bulk}

C_i

so $\frac{1}{A}\left(\frac{-dN}{dt}\right) = k_g(C_B - C_i)$

as $i = \frac{nF}{A}\left(-\frac{dN}{dt}\right)$

$\therefore i = nFk_g(C_B - C_i)$ where $k_g = D/x$

* There is a max. i at $c_i = 0$; $\quad i_L = nFk_gC_B$

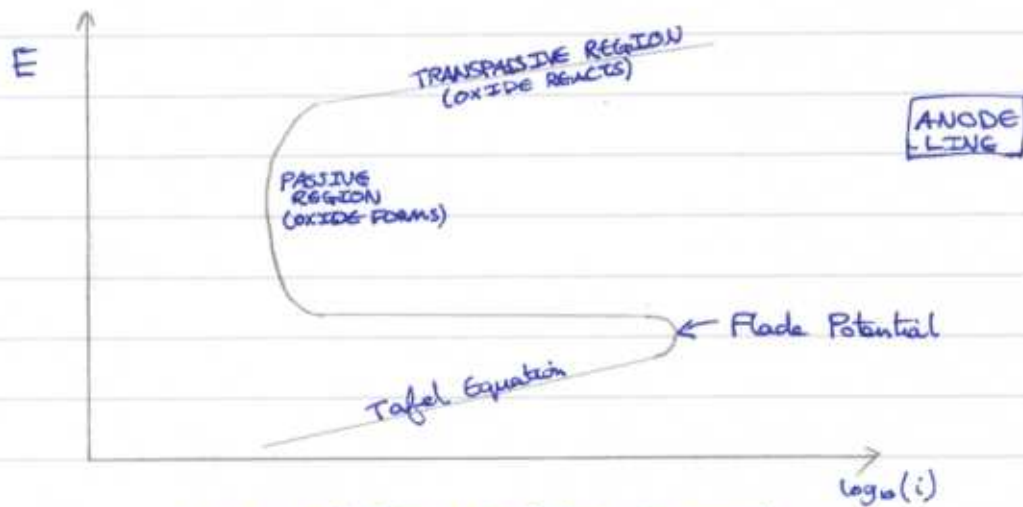* When diffusion is taken into account; $\eta_{OVERALL} = \eta_{ACTIVATION} + \eta_{CONC.}$

E

η_{ACT}

$O_2 + 2H_2O + 4e^- \rightarrow 4OH^-$

η_{CONC}

$Fe \rightarrow Fe^{2+} + 2e^-$

← change in i_a when diffusion affects anodic reaction.

$\log_{10}(i_{corr})$ $\log_{10}(i_L)$

$\log_{10}(i)$

* $E_c(\text{bulk conc.}) = E^\circ - \frac{RT}{nF}\ln\left(\frac{1}{C_B}\right)$

$E_c(\text{actual}) = E^\circ - \frac{RT}{nF}\ln\left(\frac{1}{C_i}\right)$

$1 - \frac{i}{i_L}$

$\therefore \eta \doteq E_c(\text{actual}) - E_c(\text{bulk}) = \frac{RT}{nF}\ln\left(\frac{C_i}{C_B}\right)$

Passivation ;

* This is the formation of a protective oxide layer at the surface.

EVANS DIAGRAM FOR ANODIC REACTION

* Alloying causes the curve in the above diagram to shift left.
* If other chemical species are present (e.g. Cl^- attacks passivation layers) then the transpassive line moves downwards.
* Overall corrosion behaviour ~~occurs~~ is a combination of where the anodic line intersects the possible cathodic reaction.

* Cracks in the passivating layer mean that A_a is very small and A_c is large so that corrosion is rapid.
* But e.g. Cr in Fe reacts quicker than Fe so the passivation layer of CrO_2 forms again. $Cr > 11\%$ for this to occur.